Estimation of polymer-polymer interaction density parameter by inverse gas chromatography

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The classical Flory-Huggins lattice model and Hildebrand-Scatchard solubility theory are used to develop a novel estimation scheme for determining the polymer-polymer interaction density parameter B_{23} from inverse gas chromatography data. The present considerations assume the binary interaction parameters to be concentration-dependent. It is shown that the parameter B_{23} is virtually independent of the nature of the probes. Finally, this procedure is performed on various blends of an ethylene-vinyl acetate copolymer and a chlorinated polyethylene with results substantiated by other experimental findings.

(Keywords: inverse gas chromatography; polymer-polymer interaction parameter; soluhility parameter; copolymer/polymer blends: exothermic mixings)

INTRODUCTION

Among other applications, inverse gas chromatography $(i.g.c.)$ has been widely used¹⁻⁴ to determine the polymerpolymer interaction parameter, traditionally designated by χ'_{23} . Basically, this technique monitors the retention volume V_g^0 of a volatile probe (component 1), which is carried through the chromatography column, containing a molten polyblend of polymers 2 and 3, by the continuous flow of an inert gas. The classical thermodynamic considerations of the elution behaviour of this ternary system show that the polymer-polymer interactions are indeed intimately related to the physical characteristics of the probes^{2,4}. As a result, the parameter χ'_{23} derived from i.g.c. measurements is significantly dependent on the choice of the mobile phase, thus compromising the usefulness of the technique.

Hitherto, attempts have been made to rectify this adverse situation²⁻⁶. Deshpande et al. were the first to apply the recent equation-of-state theory in connection with i.g.c. studies of plasticized poly(dimethylsiloxane) using a variety of hydrocarbons as probes². However, this model is rather sophisticated in that it requires a great deal of information, which is available only for a limited number of polymers^{7,8}. In addition, it does not warrant predictions consistent with the experimental observations. Apparently, it is simpler to characterize the polymer-polymer interactions of a particular polyblend by the average value of χ'_{23} collected from a series of probes^{3,6,9}. Since pronounced discrepancy in χ'_{23} is not uncommon, the uncertainty of this estimate may be substantial and unacceptable. The present work offers a practical solution to this problem.

METHOD OF ESTIMATION

According to the classical Flory-Huggins theory of polymer solutions¹⁰, the free energy of mixing for the above ternary system, ΔG_m , at temperature T is given by:

$$
\Delta G_{\rm m} = RT(n_1 \ln \phi_1 + n_2 \ln \phi_2 + n_3 \ln \phi_3 + n_1 \phi_2 \chi_{12} + n_1 \phi_3 \chi_{13} + n_2 \phi_3 \chi_{23})
$$
 (1)

where *R* is the gas constant, and *n*, ϕ and χ are respectively the number of moles, normalized volume fraction and binary interaction parameter, with their subscripts 1, 2 and 3 designating the component numbers cited previously. Assuming the parameters χ_{12} , χ_{13} and χ_{23} to be concentration-dependent, one obtains at infinite dilution of the probe:

$$
\left(\frac{\partial \Delta G_{\rm m}}{\partial n_1}\right)_{P,T,n_2,n_3} = RT \left[\ln \phi_1 + 1 - \frac{\overline{V}_1}{\overline{V}_2} \phi_2 - \frac{\overline{V}_1}{\overline{V}_3} \phi_3 + \phi_2 \chi_{12} + \phi_3 \chi_{13} - \frac{\overline{V}_1}{\overline{V}_2} \phi_2 \phi_3 \left(\frac{\partial \phi_3 \chi_{23}}{\partial \phi_3}\right) \right] (2)
$$

where P is the pressure and \bar{V} denotes the molar volume. It follows that, under equilibrium conditions, usually attainable at a temperature approximately 50K above the glass transition temperature, the overall polyblendprobe interaction parameter can be written as:

$$
\chi_{1(23)} = \phi_2 \chi_{12} + \phi_3 \chi_{13} - \phi_2 \phi_3 \left(\frac{\partial \phi_3 \chi_{23}'}{\partial \phi_3} \right) \tag{3}
$$

where

$$
\chi'_{23} = (\overline{V}_1/\overline{V}_2)\chi_{23} \tag{4}
$$

This parameter is readily determined by the standard i.g.c. procedure via:

$$
\chi_{1(23)} = \ln\left(\frac{273.15R(w_2v_2 + w_3v_3)}{\bar{V}_1P_1^0V_8^0}\right) - \left(\frac{B_{11} - \bar{V}_1}{RT}\right)P_1^0
$$

$$
-\left[1 - \bar{V}_1\left(\frac{\phi_2}{M_2v_2} + \frac{\phi_3}{M_3v_3}\right)\right] \tag{5}
$$

where w, v and *M* designate, respectively, the weight fraction, specific volume and molecular weight, and B_{11} and P_1^0 are the second virial coefficient and saturated vapour pressure of the probe at *T* respectively. Equation (5) is valid for $\phi_1 \simeq 0$. It may also be employed to compute χ_{12} and χ_{13} by setting $w_3=\phi_3=0$ and $w_2=\phi_2=0$ respectively. Clearly, the thermodynamic quantities $\chi_{1(23)}$, χ_{12} and χ_{13} depend considerably on the physical properties of the low-molecular-weight solvents.

Recently, it has been shown that the polymer-probe interaction parameter obtained by i.g.c, measurements conforms remarkably well to Hildebrand-Scatchard solubility theory^{$11-14$}. In this connection, we have:

$$
\bar{\chi}_{jk} = \frac{(\delta_{0j} - \delta_{0k})^2}{RT} + \bar{\gamma}_{jk} \tag{6}
$$

where δ denotes the solubility parameter, and:

$$
\bar{\chi}_{jk} = \chi_{jk}/\bar{V}_j \tag{7}
$$

$$
\bar{\gamma}_{jk} = \gamma_{jk}/\bar{V}_j \tag{8}
$$

where γ_{jk} is the entropic component of χ_{jk} . Here, the subscripts $jk = 12$, 13 and 23, whereas 0*j* and 0*k* attached to δ identify the solubility parameters of the foregoing three components by assigning j or k equal to the component number.

Substituting equations (6) – (8) into (3) yields an expression for $\chi_{1(23)}/\overline{V}_1$, which resembles the former. In the Appendix, we show that these equations can be represented by a general expression:

$$
\frac{\chi_{1(ij)}}{\bar{V}_1} = \bar{\chi}_{1(ij)} = \frac{(\delta_{01} - \delta_{ij})^2}{RT} + \bar{\gamma}_{1(ij)} \qquad ij = 02, 03, 23 \quad (9)
$$

where

$$
\delta_{ij} = \phi_i \delta_{0i} + \phi_j \delta_{0j} \tag{10}
$$

$$
\bar{\gamma}_{1(ij)} = \frac{\gamma_{1(ij)}}{\bar{V}_1} = \phi_i \bar{\gamma}_{1i} + \phi_j \bar{\gamma}_{1j} - \phi_i \phi_j \left(\frac{\partial \phi_j \bar{\gamma}_{ij}}{\partial \phi_j}\right) \tag{11}
$$

with $\phi_0 = 0$ and $\phi_i + \phi_j = 1$ indicating $\bar{\gamma}_{1(02)} = \bar{\gamma}_{12}, \bar{\gamma}_{1(03)} =$ γ_{13} and $\bar{\chi}_{1(02)} = \bar{\chi}_{12}$, $\bar{\chi}_{1(03)} = \bar{\chi}_{13}$.

Equation (9) is converted to a useful form:

$$
\frac{\delta_{01}^2}{RT} - \bar{\chi}_{1(ij)} = -\left(\frac{\delta_{ij}^2}{RT} + \bar{\gamma}_{1(ij)}\right) + \left(\frac{2\delta_{ij}}{RT}\right)\delta_{01} \qquad (12)
$$

If $\bar{\gamma}_{1(ij)}$ is not a function of δ_{01} (ref. 13), a plot of the left-hand side of equation (12) against δ_{01} would produce a straight line with intercept and gradient providing the information on $\bar{\gamma}_{1(ij)}$ and δ_{ij} respectively. The solubility parameter of solvent δ_{01} is computed by a method outlined by DiPaola-Baranyi *et al. 11.* Equation (10) can be written in an alternative form:

$$
\delta_{ij} = \delta_{02} + (\delta_{03} - \delta_{02})\phi_3 \tag{13}
$$

where $\delta_{ij} = \delta_{02}$, δ_{03} or δ_{23} depends on the value of ϕ_3 . Hence, a linear plot of δ_{ij} against ϕ_3 furnishes the estimates of δ_{02} and δ_{03} . The derivative in equation (11) is accessible, once the parameter $\bar{\gamma}_{1(23)}, \bar{\gamma}_{12}$ and $\bar{\gamma}_{13}$ have been established by equation (12). In the present analysis, we set:

$$
f(\phi_3) = \partial \phi_3 \bar{\gamma}_{23} / \partial \phi_3 \tag{14}
$$

which is treated as a polynomial empirically. Hence, we have from equations (6) , (7) and (14) :

$$
\bar{\chi}_{23} = \frac{\chi_{23}}{\bar{V}_2} = \frac{(\delta_{02} - \delta_{03})^2}{RT} + \frac{1}{\phi_3} \int_0^{\phi_3} f(\phi_3) d\phi_3 \quad (15)
$$

Equation (15) facilitates the calculation of the polymerpolymer interaction density parameter B_{23} , defined by:

$$
B_{23} = RT\bar{\chi}_{23} \tag{16}
$$

Unlike the parameter χ'_{23} , B_{23} is not related to V_1 explicitly.

RESULTS AND DISCUSSION

Walsh and coworkers⁹ have recently investigated the miscibility of an ethylene-vinyl acetate polymer (EVA 45) and a chlorinated polyethylene (CPE 3) by the i.g.c. method. Their experimental data are employed for the present purpose. The specific volumes of EVA 45 (component 2) and CPE 3 (component 3) are respectively estimated to be 1.229 and $0.817 \text{ m1} \text{ g}^{-1}$ at 343.2 K and 1.262 and $0.847 \text{ m} \text{ kg}^{-1}$ at 373.2 K. Other relevant information is detailed in ref. 9.

A typical plot constructed according to equation (12) is shown in *Figure 1* for a total of five probes in a 75/25 blend of EVA 45/CPE 3 at 343.2 K. Other systems of interest have been analysed likewise. *Table 1* lists the results on the thermodynamic parameters thus obtained for EVA 45, CPE 3 and their binary blends at 343.2 and 373.2 K.

In any event, the value of correlation coefficient r is close to unity, thus justifying the validity of equation (12). However, the foregoing linearity may result from the strong correlation between δ_{01}^2 and δ_{01} . Hence, it is necessary to examine the consistency of the observed and predicted $\bar{\chi}_{1(ij)}$ as shown in *Table 2*. The highest standard error of estimate, σ , defined in *Table 2* is 1.0×10^{-3} , which corresponds to an error of $\sim 10\%$ in $\bar{\chi}_{1(ij)}$. This value is indeed acceptable, considering the gross uncertainty in a number of parameters involved in equation (5), particularly V_8^0 , P_1^0 , M_2 and M_3 (ref. 4). Certainly, the accuracy of these measurable quantities can be greatly improved by a refined i.g.c, procedure recently proposed by E1-Hibri *et al. 12'13.* Although both polar and nonpolar probes are utilized in this connection, they do not seem to display any specific interactions, which would invalidate the foregoing analysis based on the Hildebrand-Scatchard equation, with the polymers used. This is parallel to the work reported by others^{11,14-16}, who have dealt with the polymer-probe systems exclusively. However, solvents of smaller size including acetone, methanol, chloroform and dichloromethane are found to deviate from equation (12) in the present study⁹.

Figure 1 Linear plot of $(\delta_0^2 /RT - \bar{\chi}_{1(ij)}/\bar{V}_1)$ against $\delta_{0,1}$ for the probes in a blend of EVA 45/CPE 3 with $w_3=0.25$ at 343.2K

^a Correlation coefficient

Table 2 Comparison of observed and predicted $\bar{\chi}_{1(i)}$ for EVA 45/CPE 3 systems in various probes

T(K)	No.	Probe	δ_{01} $(cal1/2 ml-1/2)$	$w_3 = 0$ $\bar{\chi}_{1(02)} \times 10^{2}$		$w_3 = 0.25$ $\bar{\chi}_{1(23)} \times 10^{2}$		$w_3 = 0.50$ $\bar{\chi}_{1(23)} \times 10^{2}$		$w_3 = 0.75$ $\bar{\chi}_{1(23)} \times 10^{2}$		$w_3 = 1.00$ $\bar{\chi}_{1(03)} \times 10^{2}$	
					Obs. ⁴ Calc. b		Obs. Calc.	Obs.	Calc.	Obs.	Calc.		Obs. Calc.
343.2		Ethyl acetate	8.13	1.17	1.21	1.30	1.31	1.35	1.35	1.23	1.14	1.22	1.14
		Tetrahydrofuran	8.22	1.09	1.24	1.19	1.33	1.23	1.37	1.07	1.15		
		Diethyl ether	6.74	1.01	1.04	1.21	1.19	1.17	1.31	1.26	1.27	1.56	1.61
	4	n-Pentane	6.32	1.11	1.10	1.26	1.27	1.52	1.41	1.43	1.42	1.91	1.86
		2-Butanone	8.24	1.41	1.25	1.49	1.34	1.56	1.52			1.07	1.12
	6	$\sigma \times 10^{2c}$	$\qquad \qquad \blacksquare$	0.10		0.09		0.10		0.06		0.06	
373.2	7	Ethyl acetate	7.89	1.38	1.57	0.92	1.02	1.41	1.44	1.10	1.19	1.29	1.33
	8	Tetrahydrofuran	8.07	1.72	1.65	1.02	1.08	1.47	1.49	1.22	1.22	1.38	1.34
	9	Diethyl ether	6.48	1.20	1.22	0.80	0.85	1.22	1.31	1.35	1.26	1.54	1.57
	10	n-Pentane	6.07	1.24	1.21	0.96	0.90	1.45	1.37	1.33	1.38	1.76	1.73
	11	2-Butanone	8.11	1.76	1.67	1.26	1.10	1.57	1.51	1.31	1.23	1.35	1.34
	12	$\sigma \times 10^{2c}$	$\qquad \qquad -$	0.10		0.095		0.06		0.07		0.03	

^a Observed $\bar{\chi}_{1(ij)}$

^b Predicted $\overline{\chi}_{1(ij)}$ by equation (9) based on the values of δ_{ij} and $\overline{\gamma}_{1(ij)}$ listed in *Table 1*

 (Δ_i^2/n) , where Δ_j is the difference between the observed and predicted $\bar{\chi}_{1(ij)}$ for the jth probe, and n is the total number of probes studied $\langle j \rangle$

Figure 2 Linear plots of δ_{ij} against ϕ_3 for EVA 45/CPE 3 at 343.2 K (\bullet) and 373.2 K (\circ)

The two sets of δ_{ij} values included in *Table 1* are plotted against ϕ_3 in *Figure 2*, which results in $\delta_{02}=6.87$ and δ_{03} = 8.43 cal^{1/2} ml^{-1/2} at 343.2 K, and δ_{02} = 6.33 and δ_{03} = 7.82 cal^{1/2} ml^{-1/2} at 373.2 K. Their respective r values are 0.9707 and 0.9879. Equation (11) leads to $f(\phi_3)$, which can be adequately expressed by the following quadratic functions:

$$
f(\phi_3)
$$
 (mol ml⁻¹) = -0.00513-0.0273 ϕ_3 +0.0486 ϕ_3 ² (17)

$$
f(\phi_3)
$$
 (mol ml⁻¹)=0.0723-0.310 ϕ_3 +0.316 ϕ_3 ² (18)

for 343.2 and 373.2 K respectively. Consequently, equations (17) and (18), respectively, render:

$$
B_{23} \left(\text{cal ml}^{-1} \right) = -1.084 - 9.27 \phi_3 + 11.05 \phi_3^2 \quad (19)
$$

$$
B_{23} \left(\text{cal ml}^{-1} \right) = 55.84 - 114.9 \phi_3 + 77.9 \phi_3^2 \tag{20}
$$

which are independent of the nature of the probe, as expected.

In *Figure 3,* curves A and B are respectively derived from equations (19) and (20) , and the data points are: based on a method employed in the original paper⁹. The latter method rests on a dubious assumption that all binary interaction parameters are independent of the concentration of the polyblend. Basically, it computes the arithmetic mean of the $\bar{\chi}_{23}$ values corresponding to the probes studied. It has been found that the standard error of y on δ_{01} , where y denotes the left-hand side of equation (12), is consistently and significantly lower than the standard deviation from the mean of $\bar{\chi}_{23}$. This implies that the full curves in *Figure 3* are more representative and reliable than the broken curves joining the data points. Indeed, our results demonstrate that mixtures of EVA 45 and CPE 3 in all proportions are exothermic at 343.2 K, due to specific interactions between the two polymers. This is positively confirmed by the phase diagram of the system, which exhibits a lower critical solution temperature significantly higher than 343.2 K^{17} . Clearly, the intensity of the foregoing physical affinity is drastically reduced upon heating, as illustrated by curve B in *Figure 3.* In fact, the complementary study cited above has reported that phase separation in EVA 45/CPE 3 blends does occur for $\bar{\phi}_3 \ge 20\%$ at 373.2 K. In contrast, the predictions of the second approach seem to suggest that exothermic mixtures are effected at both tempera-

Figure 3 Composition dependences of B_{23} for EVA 45/CPE 3 at 343.2 K (curve A) and 373.2 K (curve B). The data points are obtained from the mean of $\bar{\chi}_{23}$ for the four or five distinct probes designated by $\bar{\chi}_{23}$, via $B_{23} = RT\bar{\chi}_{23}$; 343.2K (...) and 373.2K (...). The parameter $\bar{\chi}_{23}$ is computed by $\bar{\chi}_{23} = (\phi_2 \bar{\chi}_{12} + \phi_3 \bar{\chi}_{13} - \bar{\chi}_{1(23)})/\phi_2 \phi_3$ as described in ref. 9

tures cited, but over certain ranges of ϕ_3 only, as inferred from the broken curves. Hence, the proposed algorithm is certainly more dependable in estimating the parameter B_{23} . The implication of the present finding is striking in that the conventional i.g.c, technique can now be applied effectively and routinely for the thermodynamic studies of the miscibility of polyblends.

ACKNOWLEDGEMENT

The author wishes to acknowledge the financial support of the Vote F of University of Malaya.

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APPENDIX

According to equation (7), we have:

$$
\chi_{12} = V_1 \overline{\chi}_{12} \tag{A.1}
$$

$$
\chi_{13} = \overline{V}_1 \overline{\chi}_{13} \tag{A.2}
$$

$$
\chi_{23} = \overline{V}_2 \overline{\chi}_{23} \tag{A.3}
$$

Combining equations (4) and (A.3) yields:

 χ'_{23}

$$
\chi'_{23} = \overline{V}_1 \overline{\chi}_{23} \tag{A.4}
$$

Substituting equations $(A.1)$, $(A.2)$ and $(A.4)$ into (3) , we obtain:

$$
\frac{\chi_{1(23)}}{\bar{V}_1} = \phi_2 \bar{\chi}_{12} + \phi_3 \bar{\chi}_{13} - \phi_2 \phi_3 \left(\frac{\partial \phi_3 \bar{\chi}_{23}}{\partial \phi_3} \right) \quad (A.5)
$$

Equation (6) results in:

$$
\bar{\chi}_{12} = \frac{(\delta_{01} - \delta_{02})^2}{RT} + \bar{\gamma}_{12}
$$
 (A.6)

$$
\bar{\chi}_{13} = \frac{(\delta_{01} - \delta_{03})^2}{RT} + \bar{\gamma}_{13}
$$
 (A.7)

$$
\bar{\chi}_{23} = \frac{(\delta_{02} - \delta_{03})^2}{RT} + \bar{\gamma}_{23}
$$
 (A.8)

Substituting equations
$$
(A.6)
$$
– $(A.8)$ into $(A.5)$, we obtain:

$$
\frac{\chi_{1(23)}}{\bar{V}_1} = \frac{\phi_2(\delta_{01}-\delta_{02})^2 + \phi_3(\delta_{01}-\delta_{03})^2 - \phi_2\phi_3(\delta_{02}-\delta_{03})^2}{RT}
$$

where

$$
\bar{\gamma}_{1(23)} = \phi_2 \bar{\gamma}_{12} + \phi_3 \bar{\gamma}_{13} - \phi_2 \phi_3 \left(\frac{\partial \phi_3 \bar{\gamma}_{23}}{\partial \phi_3} \right) \quad (A.10)
$$

 $+\bar{\gamma}_{1(23)}$ (A.9)

Equation (A.9) can be further simplified to:

$$
\frac{\chi_{1(23)}}{\bar{V}_1} = \frac{(\delta_{01} - \delta_{23})^2}{RT} + \bar{\gamma}_{1(23)} \tag{A.11}
$$

where

$$
\delta_{23} = \phi_2 \delta_{02} + \phi_3 \delta_{03} \tag{A.12}
$$

It can be shown that equations $(A.6)$, $(A.7)$ and $(A.11)$ are readily generalized by equation (9), setting:

$$
\bar{\chi}_{12} = \bar{\chi}_{1(02)} \tag{A.13}
$$

 $\bar{\chi}_{13} = \bar{\chi}_{1(03)}$ (A.14)

- $\bar{\gamma}_{12} = \bar{\gamma}_{1(02)}$ (A.15)
	- $\bar{y}_{13} = \bar{y}_{1(03)}$ (A.16)